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# Lithium Orthosilicate-Linear Low-Density Polyethylene Nano-Composite as Substrate for High Frequency Devices: Dielectric Characterization

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Dielectric properties of nanosized Lithium Orthosilicate (Li<sub>4</sub>SiO<sub>4</sub>) and linear low-density polyethylene composite at X-Band frequencies are investigated, to realize its utility as a substrate for microwave devices. Solid state technique is used to synthesize nano Li<sub>4</sub>SiO<sub>4</sub>. To confirm the structural and morphological properties of the synthesized nano Li<sub>4</sub>SiO<sub>4</sub>, X-ray diffraction, Fourier transformed infrared spectroscopy and transmission electron microscopy are used. Composites of nano Li<sub>4</sub>SiO<sub>4</sub> and linear low-density polyethylene matrix are synthesized with varying weight fraction of nano Li<sub>4</sub>SiO<sub>4</sub> viz. 2%, 4% and 6%. The lateral portion of a fractured sample (composite) is examined using scanning electron microscope to confirm homogeneous dispersion of nano Li<sub>4</sub>SiO<sub>4</sub> in linear low-density polyethylene matrix. Water absorption measurement of all the composites are carried out based on ASTM D570-98. Densities of the composites are measured via hydrostatic weighing by using Archimedes principle. Complex permittivity and permeability measurement of the composites are recorded by utilizing Nicholson-Ross approach. For all weight fraction of the nano inclusions, the real component of permittivity and dielectric loss tangent of the composites in the X-band are found to be in the range of 2.2 – 2.6 and 10<sup>2</sup> – 10<sup>4</sup>, respectively. Return losses for the composites are calculated from complex permittivity and permeability values to validate its applicability as substrate for various high frequency applications.

Keywords: Complex permittivity; Complex permeability; Return loss; X-band

# 1 Introduction

With the growth of electronic industries, the demand of tailored substrates for high frequency devices has increased manifolds. The complex permittivity and permeability of any material determines its suitability as substrates for these devices. A single material substrate may not always be able to satisfy all the requirements for a certain application. Various composites, such as inclusion filled polymer matrix (nano, ceramic, dielectric magnetic, *etc.*) can be a viable choice. Ceramics have high dielectric strength, moderate dielectric constant and low loss tangent. Hence, dielectric ceramic nano materials and polymer composites can be regarded as feasible microwave substrate materials for high frequency application.

Many ceramic nanomaterials, including Mg<sub>2</sub>SiO<sub>4</sub>, Ba(Mg<sub>1/3</sub>Ta<sub>2/3</sub>)O<sub>3</sub>, Mg<sub>4</sub>Nb<sub>2</sub>O<sub>9</sub>, ZnAl<sub>2</sub>O<sub>4</sub>, *etc.*, have been studied for their interesting dielectric properties in microwave frequencies. Cerium oxide (CeO<sub>2</sub>) filled polytetrafluoroethylene (PTFE) is synthesized for microwave substrate applications, using powder

measured value of the real part of complex permittivity ( $\varepsilon'$ ) is 5 and dielectric loss tangent ( $\tan \delta_e$ ) is 6.4×10<sup>-3</sup> respectively at 7GHz. Dielectric properties of PTFE loaded with varying %VF of MgTiO<sub>3</sub> is studied.<sup>5</sup> The measured value of  $\varepsilon'$  is 5.5 and  $\tan \delta_e = 2.7 \times 10^{-4}$  (at 10 GHz) makes the composite, a promising material for electronic packaging. Dielectric properties with varying weight fraction (%WF) of titania (TiO<sub>2</sub>) filled low density polyethylene (LDPE) composite is studied as a microstrip patch antenna (MPA) substrate in the X-band. The composite with 6% WF shows an increase in  $\varepsilon'$  from 2.13 to 2.5 with increasing frequency. The reported value of  $\tan \delta_e$  is  $4 \times 10^{-3}$ making it a suitable substrate material for MPA. The discussed composite is used as a graded substrate, which as a result enhances the bandwidth and S11 of the antenna. A comparative investigation of dielectric behaviour of barium titanate (BaTiO<sub>3</sub>) in X-band with

polyaniline (PANI) and maleic resin as polymer

processing technique by increasing CeO2 volume

fraction (%VF) up to 0.6%. Cavity perturbation

technique is used to measure the microwave dielectric

characteristics. For 0.6 %VF of filler loading the

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matrix is reported. It is found that a very small amount of BaTiO<sub>3</sub> in insulating PANI reduces the value of  $\varepsilon'$  and  $\tan \delta_e$  up to 26 and 0.2. While in case of BaTiO<sub>3</sub>-maleic resin the value of ε' and tanδ<sub>e</sub> decreases gradually with decreasing filler concentration and reached a minimum value of 2.3 and 0.3 respectively. X-band dielectric property of PTFE-TiO<sub>2</sub> is studied by cavity perturbation method by using nano sized and micron sized TiO<sub>2</sub>. In the studies it is observed that  $\varepsilon'$  of nano sized composite is higher than the micron sized composite. It is also observed that  $\epsilon'$  and  $tan\delta_e$  increases with %WF and reaches maximum value of 7 and 10<sup>-2</sup> for 50%WF.<sup>8</sup> A comparative dielectric study is done for PTFE matrix loaded with alumina (Al<sub>2</sub>O<sub>3</sub>) and magnesium oxide (MgO) in the X band for microwave substrate application. From investigation it is found that PTFE-Al<sub>2</sub>O<sub>3</sub> composite has a  $\varepsilon'$  of 4.30 and  $\tan \delta_{\varepsilon}$  of  $2.1 \times 10^{-3}$  whereas  $\varepsilon'$ and tanδ<sub>e</sub> of the PTFE-MgO composite is 3.35 and 1.5×10<sup>-3</sup> respectively for optimum filler loading. Composite of barium strontium titanate (BST) and thermoplastic cyclic olefin copolymer is investigated with different loadings of BST, using bulk and nano sized ceramic powders for high frequency application. The value of  $\varepsilon'$  and  $\tan \delta_{\varepsilon}$  is measured as a function of filer loading at 1GHz. The result shows moderate value of  $\varepsilon' = 6$  and low value of  $\tan \delta_e = 9 \times 10^{-3}$ . 10 Dielectric properties of CaTiO<sub>3</sub> loaded polypropylene composite fabricated through compression molding method are studied by using x-band cavity perturbation technique for microwave substrate application. The composite have an effective value of  $\varepsilon'$  and  $\tan \delta_e$  is 11.74 and  $7 \times 10^{-3}$  respectively for highest filler concentration. 11 Dielectric properties of polystyrene -magnesium based ceramics (MgTa<sub>2</sub>O<sub>6</sub>, MgNb<sub>2</sub>O<sub>6</sub>, and MgWO<sub>4</sub>) is investigated in the frequency range 1–7.3 GHz frequency for microwave substrate and electronic packaging application. The value of  $\varepsilon'$  is nearly constant reporting 25.48, 20.84 and 12.44 respectively for MgTa<sub>2</sub>O<sub>6</sub>, MgNb<sub>2</sub>O<sub>6</sub>, and MgWO<sub>4</sub> with tanδ<sub>e</sub> nearly 10<sup>-3</sup> for all the materials. 12 A coplanar waveguide monopole antenna is fabricated over  $Ca[(Li_{1/3}Nb_{2/3})_{0.8}Ti_{0.2}]0_{3-8}$  (CLNT)epoxy composite. From the dielectric investigation it is observed that value of  $\varepsilon'$  increases with increase in CLNT concentration. The reported value of  $\epsilon'$  and  $\tan \delta_e$  for 0.4 %VF is 7.7 and  $4 \times 10^{-3}$  respectively at 9 GHz.<sup>13</sup> High-density polyethylene (HDPE) and BaO-Nd<sub>2</sub>O<sub>3</sub>-TiO<sub>2</sub> (BNT) coated with LDPE

composite is synthesized and are used for microwave substrate applications. The composites  $\varepsilon'$  value decreased from 11.87 (at 7 GHz) to 3.45 (at 9 GHz), while  $\tan \delta_e$  remain below  $1.6 \times 10^{-3.14}$  Dielectric resonator structures for wearable devices on cyclic olefin copolymer (COC) and barium strontium titanate (BST) composites are studied at 2.45 GHz frequency. The  $\epsilon'$  and  $tan\delta_e$  of BST- COC's are 14.5 and 2.35 x 10<sup>-3</sup>, respectively. 15 A multi-layer aluminaceramic and polyimide substrate active resonating structure is proposed for wireless communication systems. At 10 GHz, the device is fabricated on a four-layer alumina-ceramic substrate having  $\varepsilon' = 9.0$ and  $tan\delta_e = 10^{-3}$ , as well as a four-layer polyimide substrate with  $\varepsilon' = 3.2$  and  $\tan \delta_e = 2 \times 10^{-2.16}$  It is seen that, substrates with ceramic included in a polymer matrix with varying %WF can be utilized for various high frequency devices. This report has exercised this fact, and investigated to fill the research gap related to selection of suitable ceramic material as fillers.

LLDPE is a polymer with linear qualities and acceptable electrical properties, as well as good tensile properties, chemical resistance, low water absorption, and high thermal stability. decomposition temperature of LLDPE is 438.85 °C with weight loss of 0.01%. The major weight loss occurs in between the temperatures 441.98 and 513.15 °C due to growth of volatiles 17-19. Silicates are recognized for their excellent dielectric characteristics, low loss tangent and minimal thermal expansion.<sup>20</sup> Chemical, physical, mechanical and magnetic properties of nano ceramics differ from other materials like metals and conventional bulk ceramic materials, this makes it a possible candidate for this investigation.

In this work, nano Li<sub>4</sub>SiO<sub>4</sub> is synthesized and are used as inclusions in a linear low-density polyethylene (LLDPE) polymer matrix with varying %WF (viz. 2%, 4%, and 6%) to investigate its high frequency dielectric properties. Along with other structural and morphological studies, X band complex permittivity and permeability of composites (LLDPEnano Li<sub>4</sub>SiO<sub>4</sub>) is measured using Nicolson Ross technique.

# 2 EXPERIMENTAL WORKS

2.1 Synthesis of nanosized  $\text{Li}_4\text{SiO}_4$  and its composite with LLDPE

Nanosized lithium orthosilicates are synthesized by following conventional solid-state method by mixing

high purity ingredients of nanosized silica dioxide (SiO<sub>2</sub>) and lithium carbonate (Li<sub>2</sub>CO<sub>3</sub>) (Sigma Aldrich) in stoichiometric proportions. Nano Li<sub>2</sub>CO<sub>3</sub> and SiO<sub>2</sub> are mixed in 2:1 molar ratio in an agate mortar for 1 hour. The mixture is then calcined at 900 °C for 4 hour.21 To make the composite, LLDPE is dissolved in toluene and stirred with a mechanical stirrer at 100 °C. After dissolving the synthesized Li<sub>4</sub>SiO<sub>4</sub> nano particles are added in WF increments of 2%, 4% and 6% to make a homogeneous mixture of the inclusions in the polymer. When the toluene is completely evaporated, a viscous solution is formed. The solution is then poured into a die with dimensions of 10 x 22 x 2 mm<sup>3</sup>, allowed to congeal for 2 hours at room temperature, and a composite substrate of LLDPE-nano Li<sub>4</sub>SiO<sub>4</sub> is obtained.<sup>22</sup>

# 2.2 Structural and Morphological Characterization

The X-ray diffraction (XRD) patterns of nano Li<sub>4</sub>SiO<sub>4</sub> is recorded using a Bruker D8 (Make: BRUKER AXS, GERMANY, Model: D8 FOCUS) X-ray powder diffractometer with Cu Kα radiation  $(\lambda = 1.5406)$ . Fig. 1 shows the XRD patterns of Lithium Orthosilicate nano particles. The peaks at  $2\theta = 16.79^{\circ}$ , 22.18°, 24.17°, 29.27°, 34.07o, 36.86°, 39.56°, 48.75°, 63.33 ° and 73.92° corresponding to the (h k l) value  $(1\ 0\ 0), (\bar{1}\ 1\ 0), (1\ 0\ 1), (0\ 2\ 0), (0\ 2\ 1), (\bar{2}\ 1\ 0), (\bar{1}\ 0\ 2),$  $(\overline{2}\ 2\ 1)$ ,  $(0\ 4\ 1)$   $(\overline{2}\ 4\ 1)$ , are the diffraction peaks of Lithium Orthosilicate (ICDD 00-037-1472). The sample has monoclinic structure, with lattice parameters of a = 5.39, b = 9.39, and c = 4.66. Along with the dominant Li<sub>4</sub>SiO<sub>4</sub> peaks, some peaks of Li<sub>2</sub>CO<sub>3</sub> are also observed due to incomplete reaction. The average crystallite size is calculated using Debye-Scherrer formula<sup>23</sup> and is found to be 150 nm.

FTIR spectra in the range of 400–4000 cm<sup>-1</sup> are obtained using a NICOLET FTIR spectrometer.

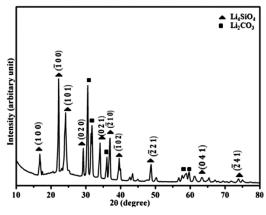


Fig. 1 — XRD of nano Li<sub>4</sub>SiO<sub>4</sub>.

(Model: IMPACT 410) (Make: NICOLET). For this the material is mixed with KBr and crushed to a very fine powder and in a moisture-free environment, a transparent disc is created to obtain FTIR spectra. The FTIR spectrum of the synthesized Li<sub>4</sub>SiO<sub>4</sub> is shown in Fig. 2. The -OH stretching vibration caused by physical adsorption of water and bound water is responsible for the broad characteristic band ranging from 3770 – 2480 cm<sup>-1</sup>. The distinctive bands observed from FTIR measurements at 1186-1047 cm<sup>-1</sup> and 480 cm<sup>-1</sup> correspond to Si–O–Si bending vibrations and O–Li–O stretching vibrations, respectively. The band observed around 862 cm<sup>-1</sup> and 1423 cm<sup>-1</sup> is due to the presence of carbonates in the form of Li<sub>2</sub>CO<sub>3</sub>.

Transmission electron microscope (TEM) (Model: TECNAI G2 20 S-TWIN)(FEI COMPANY, USA) is used to confirm the average particle size of the synthesized Li<sub>4</sub>SiO<sub>4</sub>. According to the TEM image in Fig. 3, the average particle size of nano Li<sub>4</sub>SiO<sub>4</sub>

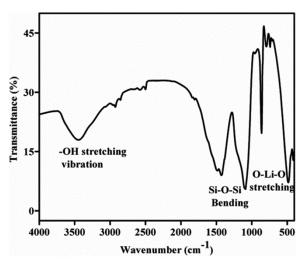


Fig. 2 — FTIR of nano Li<sub>4</sub>SiO<sub>4</sub>.

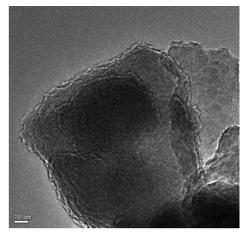


Fig. 3 — TEM of nano Li<sub>4</sub>SiO<sub>4</sub>.

complements the calculated value obtained from Debye Scherrer formula. The lateral side of cracked samples of 6% WF is examined using a scanning electron microscope (SEM)(Model: JSM- 35CF, Make: JEOL) to validate the uniform dispersion of Li<sub>4</sub>SiO<sub>4</sub> in LLDPE composite substrates (Model: JSM- 35CF, Make: JEOL). Fig. 4 shows SEM images of LLDPE-nano Li<sub>4</sub>SiO<sub>4</sub> composite substrates. In some portions the agglomeration of nanomaterials has occurred during mixing due to high viscosity of LLDPE.

Adverse environmental circumstances, such as excessive moisture and temperature, can cause material characteristics to alter. So properties such as water absorbance, density, and coefficient of thermal expansion (CTE) must be investigated. Water absorption is measured according to ASTM D570-98, and the changing trend of percentage water absorption with time of immersion is shown in Fig. 5.<sup>24</sup>

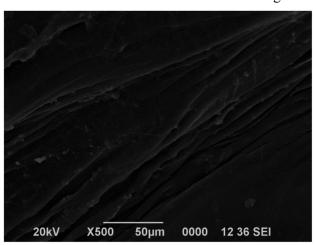


Fig. 4 — SEM of LLDPE- nano Li<sub>4</sub>SiO<sub>4</sub> (6%) composite.

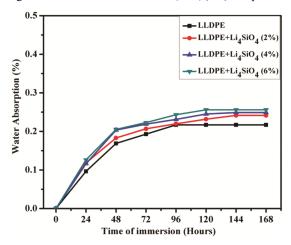


Fig. 5 — Water absorption of pure LLDPE and LLDPE- nano Li<sub>4</sub>SiO<sub>4</sub> composite.

To measure the water absorption sample of size  $10\times22\times2$  mm<sup>3</sup> is immersed in water and weight is measured in a weighing balance of precision 0.0001 g after every 24 hr of immersion until saturation is achieved. The saturation is achieved after 168 hr of immersion. From the recorded weight the percentage of water absorption after each 24 hr of immersion is calculated by using equation (1) and the calculated values are given in Table 1. From the table it is seen that for the 6 %WF the saturation water absorption is only 0.25%.

$$w_a(in \%) = \frac{w_f - w_i}{w_i} \times 100$$
 ... (1)

Where

w<sub>a</sub>=Percentage of absorbing water.

W<sub>f</sub>= Final weight of the pieces.

W<sub>i</sub>= Initial weight of the pieces.

The densities of the composites are estimated by using Archimedes principle utilizing hydrostatic weighing.<sup>25</sup> For this, a half filled beaker with ethanol is weighted and the combined mass of beaker and ethanol is recorded (w<sub>1</sub>). After that composite sample is suspended in ethanol and a fresh reading is taken (w<sub>2</sub>). Finally, the sample is allowed to settle at the bottom of the beaker and reading of the combined mass is measured (w<sub>3</sub>). The density is calculated using equation (2) by using the recorded values of w<sub>1</sub> w<sub>2</sub> w<sub>3</sub> and d (ethanol density).

$$\rho = \frac{w_3 - w_1}{w_2 - w_1} \times d \ (g/cm^3) \qquad \dots (2)$$

The obtained value of density is listed in Table 2.

The coefficient of thermal expansion (CTE) is an important feature that needs to be taken into account. The mixing rule states that the CTE of a two-phase composite is directly proportional to the filler volume percent.<sup>26</sup> The composites CTE are calculated as follows in the mixing rule:

$$\alpha_{\rm c} = f \alpha_{\rm f} + (1 - f) \alpha_{\rm m} \qquad \dots (3)$$

Table 1 — Water absorbance data for the LLDPE- nano  $\text{Li}_4 \text{SiO}_4$  composite substrate Sample Water Absorption (in %) LLDPE 0.21 LLDPE+  $\text{Li}_4 \text{SiO}_4$  (2%WF) 0.23 LLDPE+  $\text{Li}_4 \text{SiO}_4$  (4%WF) 0.24 LLDPE+  $\text{Li}_4 \text{SiO}_4$  (6%WF) 0.26

Table 2 — Density of LLDPE- nano Li <sub>4</sub> SiO <sub>4</sub> composite substrate	
Material	CTE (k <sup>-1</sup> )
LLDPE	18×10 <sup>-5</sup>
$\text{Li}_4 \text{SiO}_4$	1.428×10 <sup>-5</sup>
LLDPE+ $Li_4SiO_4$ (2%)	17.6696×10 <sup>-5</sup>
LLDPE+ $Li_4SiO_4$ (4%)	17.3371×10 <sup>-5</sup>
IIDPF+Ii.SiO.(6%)	17 0056×10 <sup>-5</sup>

Where  $\alpha_f$   $\alpha_m$  and  $\alpha_c$  are the filler, matrix and composite, CTEs, respectively, and f is the filler volume percent. Table. 3 shows the computed CTE values. From the table it is seen that the change in CTE due to addition of inclusion is minimal up to the %WF of this work.

#### 2.3 Dielectric Characterization

determine complex permittivity and permeability at X-band frequency, the Nicholson-Ross method is employed.<sup>27</sup> The measurement setup includes an Agilent WR-90 X11644A rectangular waveguide line, an Agilent E8362C vector network analyzer, a sample container with a thickness of  $\lambda/4$ , and an interfacing computer. To decrease errors, the system is calibrated prior to measurements using the thru-reflect-line (TRL) technique.<sup>28</sup> Following TRL calibration, LLDPE-nano Li<sub>4</sub>SiO<sub>4</sub> composites of dimension 10×22×2 mm<sup>3</sup> are inserted into the sample holder to measure complex permittivity permeability.

Figure 6 (a&b) illustrates the  $\varepsilon'$  and  $\tan \delta_e$  of LLDPE-nano Li<sub>4</sub>SiO<sub>4</sub> for 2, 4 and 6 %WFs at X-band frequencies at room temperature. The values for pure LLDPE and Li<sub>4</sub>SiO<sub>4</sub> are 2.2 and 6.11, respectively.<sup>29</sup> The standard error of  $\varepsilon'$  and  $\tan \delta_e$  from its mean is depicted as the error bars in the plots of Fig. 6(a&b). The deviations observed can be due to various reasons such as, casting pressure and formation of

 $\label{eq:coefficient} Table \ 3 - Calculated \ Coefficient \ of \ thermal \ expansion \ of \ LLDPE- \ nano \ Li_4SiO_4 \ composite \ substrate.$ 

7 7	1
Sample	Density(g/cm <sup>3</sup> )
LLDPE	0.92
$LLDPE+ Li_4SiO_4 (2\%)$	0.95
$LLDPE+Li_4SiO_4$ (4%)	0.98
LLDPE+ Li <sub>4</sub> SiO <sub>4</sub> (6%)	1.01

agglomerates of the inclusions in the polymer matrix over the X-band frequency range, the  $\varepsilon'$  is almost constant. The average real permittivity values of the composite with 2%, 4%, and 6% inclusion content are 2.33, 2.48, and 2.55, respectively. The permittivity of the composite substrate shows an incremental trend, as the inclusion content increases. This may be due to inherent electric dipole polarization of Li<sup>+</sup> ions and interfacial polarization due to LLDPE. The presence of Li<sup>+</sup> ions causes polarization in Li<sub>4</sub>SiO<sub>4</sub>, as they are strongly polarizable. This increases the Li<sup>+</sup> ions concentration, eventually contribute to the increases in overall permittivity.<sup>30</sup> In the complex permittivity investigations, a non-linear behaviour is also observed, which is related to space charge accumulation at the various interfaces created as a result of the heterogeneous composite system of dispersant Li<sub>4</sub>SiO<sub>4</sub> in the LLDPE matrix.<sup>31</sup>

The tanδ<sub>e</sub>, of the composite LLDPE-nano Li<sub>4</sub>SiO<sub>4</sub> is shown in Fig. 5(b). Dielectric loss is primarily seen due to two types of energy dissipation: conduction loss and dielectric loss. The  $tan\delta_e$  values for the composite LLDPE-nano Li<sub>4</sub>SiO<sub>4</sub> vary between 10<sup>-2</sup> and  $10^{-4}$  with frequency. Negative values of  $\tan \delta_e$  in the X-band are seen due to the coupling impact of the host matrix and inclusion in the Fabry-Perot resonance (FPR) region.<sup>32</sup> Negative loss tangent values are not always an inherent feature of the material, but they are also affected by the thickness of the substrate.<sup>33</sup> If the substrate thickness is  $\lambda/2$ , the corresponding FPR may easily alter the connected electromagnetic field and in this work, the sample holder thickness is  $\lambda/4$ . The unusual behaviour of materials negative loss tangent is explained by Axelrod et al.<sup>34</sup> The negative loss phenomenon is

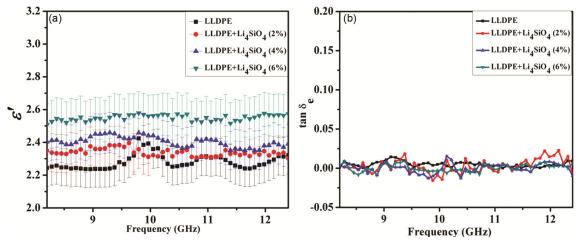


Fig. 6 — (a) Real part of complex permittivity and (b) Dielectric loss tangent of LLDPE- nano Li<sub>4</sub>SiO<sub>4</sub> composites.

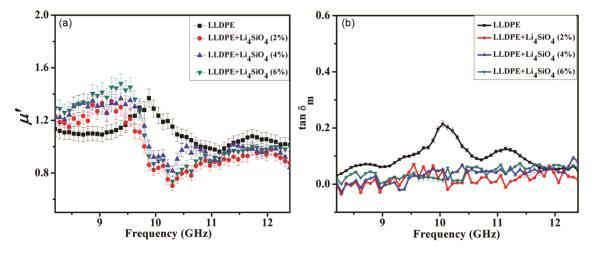


Fig. 7 — (a) Real part of complex permeability and (b) Magnetic loss tangent of LLDPE- nano Li<sub>4</sub>SiO<sub>4</sub> composites.

caused by non-compensated matrix-anchored charges inside the polymer matrix and on the interface. Non-bonding orbital's, whether full or empty for positive or negative ions, can cause charge this anchoring. As a result, the molecules reach a metastable state, in which charge separation occurs with energy accumulation. At the suitable frequency, this metastable state can be dissolved, yielding a charge avalanche recombination, resulting in energy release. When this emitted energy exceeds the absorbed, at a specific frequency negative loss phenomenon can be seen.

Figure 7 (a&b) illustrate the real part of complex permeability ( $\mu$ ') and loss tangent (tan  $\delta_m$ ) for 2%, 4%, and 6% weight fractions of LLDPE- nano Li<sub>4</sub>SiO<sub>4</sub> composite. Increasing the inclusion content in the host matrix lowers the average value of from 1.06 to 1.02. This is due to the presence of nonmagnetic Li<sub>4</sub>SiO<sub>4</sub> in the nonmagnetic polymer host matrix, which reduces intergranular magnetic interaction. The value of tan  $\delta_m$  fluctuates between  $10^{-1}$  and  $10^{-4}$ .

#### 3 Return Loss

As the complex dielectric properties are dependent on frequency, role of substrate becomes significant in any high frequency device.<sup>35</sup> The relevance of a material for a specific application such as resonator, absorber etc depends on the value of return loss of the material. A low return loss indicates that the material is suitable for absorber applications and on the contrary a high value suggests its suitability for resonator applications (viz. MPAs, dielectric resonator antennas etc). Return losses of pure LLDPE and composites are calculated by using equation (4)

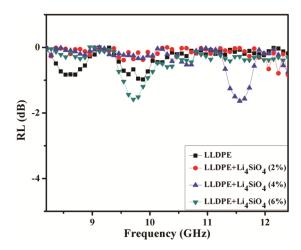


Fig. 8 — Simulated Return Loss of LLDPE- nano  $Li_4SiO_4$  composites.

$$RL(dB) = 20log_{10} \left| \frac{Z_{in} - 1}{Z_{in} + 1} \right|$$
 ... (4)

Where,

$$Z_{in} = \sqrt{\frac{\epsilon_r}{\mu_r}} tanh \left[ j \left( \frac{2\pi f d}{c} \right) \sqrt{\epsilon_r \mu_r} \right] \qquad ... (5)$$

Where  $Z_{in}$  is the input impedance in terms of free space, f is the frequency of operation, d=2 mm is the substrate thickness and  $c = 3 \times 10^8 \text{m/s}$  is the speed of light.<sup>36</sup> The calculated return loss of the composite substrate over the X-band is shown in Fig. 8.

From the figure it is seen that the overall value of return loss for all the composites are lower than -2 dB, suggesting its applicability as substrate for resonators rather than absorber. The estimated RL values complement the complex permittivity values as well. According to the literature, nanocomposites with an RL value of -45dB are suitable for absorber applications.<sup>37</sup>

#### **4 Conclusions**

Li<sub>4</sub>SiO<sub>4</sub> nano particles are synthesized following a standard solid state technique. The formation of Lithium orthosilicate is confirmed by diffraction peaks obtained in the XRD spectra. The Li-O-Li stretching vibrations and Si-O-Si bending vibrations observed from FTIR measurements also confirms the same. The TEM images reveal the nanoparticles average size as ~150 nm. The SEM images show that the inclusions in LLDPE are evenly distributed. The complex permittivity exhibits a linear trend with low fluctuations which can be attributed to buildup of space charge at the inclusion-polymer interfaces. The  $\epsilon'$  and  $\tan \delta_e$  are in the range of 2.2 – 2.6 and 10<sup>-2</sup>—10<sup>-4</sup> respectively. The study reveals that, the composite can be used as a substrate material for various devices such as microstrip patch antenna, dielectric resonators etc. The moderate values of  $\varepsilon'$ , and the computed RL values complements the same.

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